

$$\left. \begin{aligned} K_I &= \sqrt{2a} \sigma \cos^2 \theta \\ K_{II} &= \sqrt{2a} \tau \sin \theta \cos \theta \end{aligned} \right\} \rightarrow \frac{K_{II}}{K_I} = \tan \theta$$

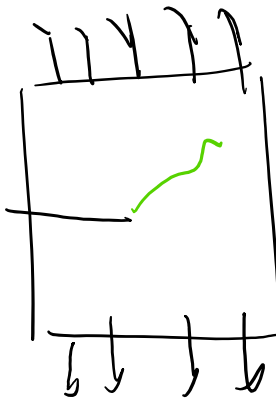
$$\theta = 30^\circ \quad \tan \theta = \frac{1}{\sqrt{3}} = .577$$

For Max. crack: criterion crack propagates when

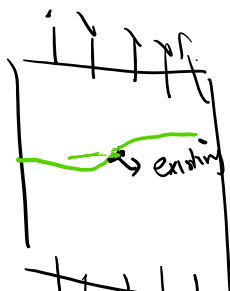
$$\frac{K_I}{K_{IC}} = .75 \rightarrow \sqrt{2a} \sigma \cos^2 30 = .75 K_{IC} \rightarrow \sigma_{max} = \frac{.75}{\frac{3}{4}} \frac{K_{IC}}{\sqrt{2a}} \rightarrow \frac{K_{IC}}{\sqrt{2a}}$$

$$MERR \quad \frac{K_I}{K_{IC}} = .835 \rightarrow \sigma_{max} \quad MERR$$

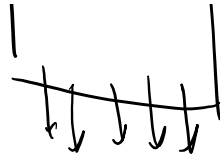
Crack nucleation criterion:



crack
Nucleation



crack that is not explicitly modeled
or defect



We can formulate appropriate crack nucleation models for each of the three crack propagation criteria:

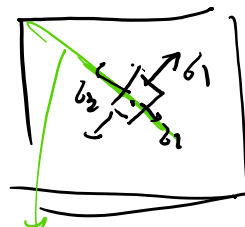
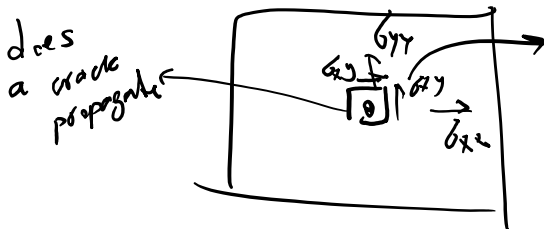
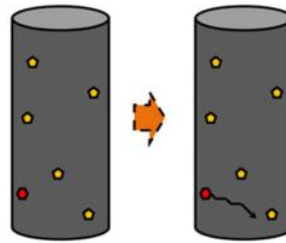
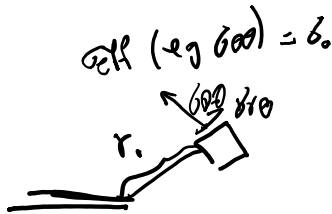
- Cracks nucleate from microscopic material defects under high stress/ strain loads.
- For each crack propagation criterion there can be a corresponding nucleation criterion.
- For example for **maximum circumferential tensile stress**, a crack nucleates when the maximum principle stress σ_1 at a point reaches material strength σ_0 :

$$\max_{-\pi < \theta < \pi} \sigma_\theta(r \rightarrow 0^+, \theta) = \sigma_1 = \sigma_0, \quad \text{crack nucleates}$$

Although we assume that there is no initial crack tip, we can measure r relative to the potential nucleation point.

- Same concept applies to **modified maximum circumferential tensile stress** criteria:

$$\max_{-\pi < \theta < \pi} \sigma_{\text{eff}}(r \rightarrow 0^+, \theta) = \sigma_0, \quad \text{crack nucleates}$$



$\sigma_1 \geq \sigma_0$
a crack is nucleated

early crack propagation direction

A nucleation model that can go well with MERR

Crack nucleation criterion

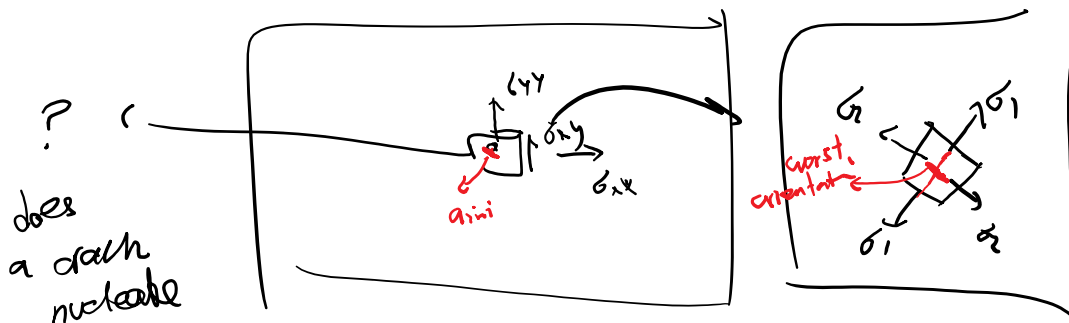
- For **Maximum Energy Release Rate Criterion** if we assume there are no defects, there will be no crack nucleation. However, assuming that local stress field generates a tensile maximum principal stress of σ_1 a "microscopic" initial crack (defect) of length a_{ini} perpendicular to σ_1 direction generates,

$$G = \frac{K_I^2 + K_{II}^2}{E'} = \pi a_{ini} \sigma_1^2$$

so the microcrack propagates (i.e., a "macroscopic" crack nucleates) when,

$$G = G_c \Leftrightarrow \sigma_1 = \sqrt{\frac{G_c}{\pi a_{ini}}}$$

- Initial crack direction perpendicular to σ_1 is chosen to maximize G .
- We have assumed the initial crack to be small enough to use the infinite domain SIF formula of $K_I = \sqrt{\pi a} \sigma$.



$$K = \sigma_1 \sqrt{\pi a_{ini}}$$

large domain approximation

what value to choose for a_{ini} ?

- SAM, NDE length scale of crack that can be detected

OR
 - a length below which we don't directly resolve material defects

$$G = \frac{K_I^2}{E'} = \frac{\sigma_1^2 \pi a_{ini}}{E'} = G_c$$

$$\rightarrow \sigma_1 \geq \sqrt{\frac{E' G_c}{\pi a}}$$

a crack can be nucleated of size a_{ini} normal to σ_1 direction

8. Fatigue

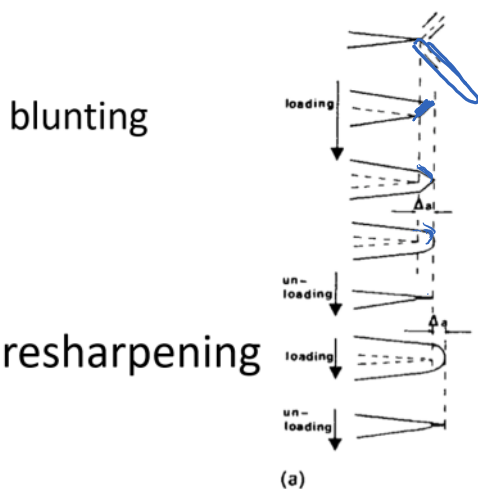
- 8.1. Fatigue regimes
- 8.2. S-N, P-S-N curves
- 8.3. Fatigue crack growth models (Paris law)
 - Fatigue life prediction
- 8.4. Variable and random load
 - Crack retardation due to overload

Fatigue happens when the applied loads cycle in time

Fatigue fracture is prevalent!

- Deliberately applied load reversals (e.g. rotating systems)
 - Vibrations (machine parts)
 - Repeated pressurization and depressurization (airplanes)
 - Thermal cycling (switching off electronic devices)
 - Random forces (ships, vehicles, planes)
- (source: Schreurs fracture notes 2012)

Fatigue occurs always and everywhere and is a major source of mechanical failure

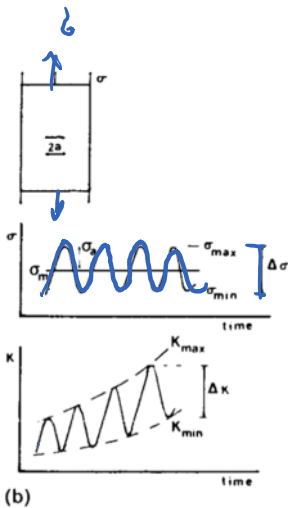
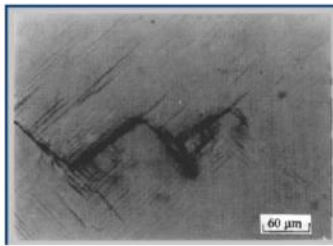
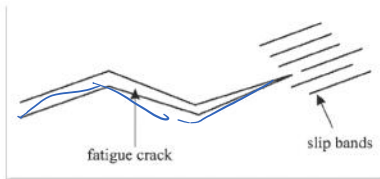




Fracture surface of a 2024-T3 aluminum alloy (source S. Suresh MIT)

incremental crack propagation
 - Each time after sharpening crack propagates a little and goes through blunting & sharpening again

Striation caused by individual microscale crack advance incidents



$K = \sqrt{\pi a} \sigma$
 For a given a $\Delta K = \sqrt{\pi a} \Delta \sigma$
 $\Delta \sigma$ is fixed
 $\Delta K \uparrow$ because $a \uparrow$
 $(\sqrt{\pi a} \Delta \sigma)$
 $\frac{a_1}{a_2} > \frac{\sigma_1}{\sigma_2}$

Types of fatigue:

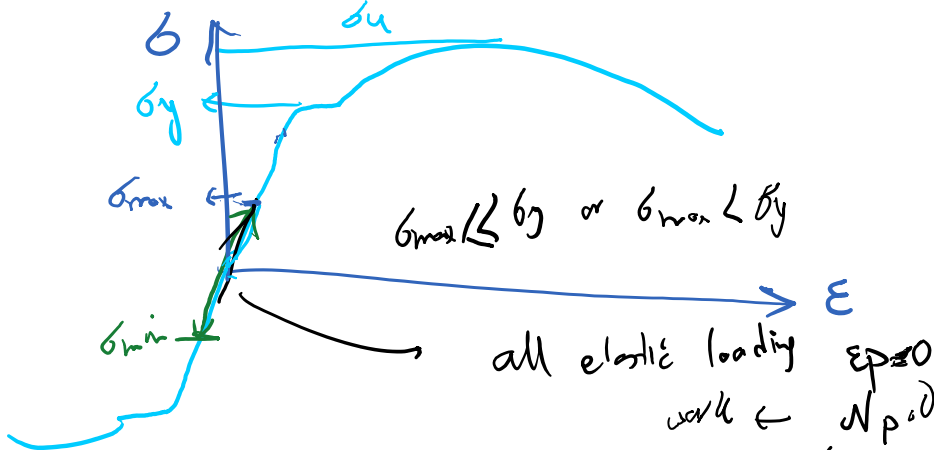
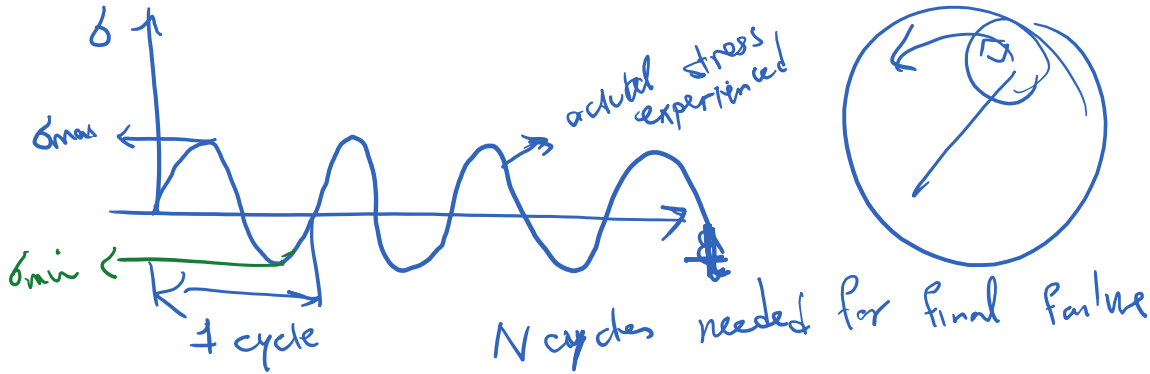
Table 7.1 Classification of fatigue damage

Fatigue	Failure cycles N_R	Pertinent stress	Strain ratio $\Delta \epsilon^p / \Delta \epsilon^e$	Energy ratio $\Delta W^p / \Delta W^e$
Very high cycle fatigue	$> 10^7$	$< \sigma_r$	≈ 0	≈ 0

Fatigue	Failure cycles N_R	Pertinent stress	Strain ratio $\Delta \epsilon^p / \Delta \epsilon^e$	Energy ratio $\Delta W^p / \Delta W^e$
Very high cycle fatigue	$> 10^7$	$< \sigma_F$	≈ 0	≈ 0
High cycle fatigue	10^5 to 10^6	$< \sigma_Y$	≈ 0	≈ 0
Low cycle fatigue	10^2 to 10^4	σ_Y to σ_U	1 to 10	1 to 10
Very low cycle fatigue	1 to 20	$\approx \sigma_U$	10 to 100	10 to 100

Source: Dufailly and Lemaitre (1995)

of cycles before final failure is in the order of M to 10 or higher M s.



- HCF, V HCF
- $N \sim 10^6 - 10^7$
 - Almost all elastic deformation
 - LEFM theory is appropriate
 - Stress (profile/history) is generally used to study fatigue

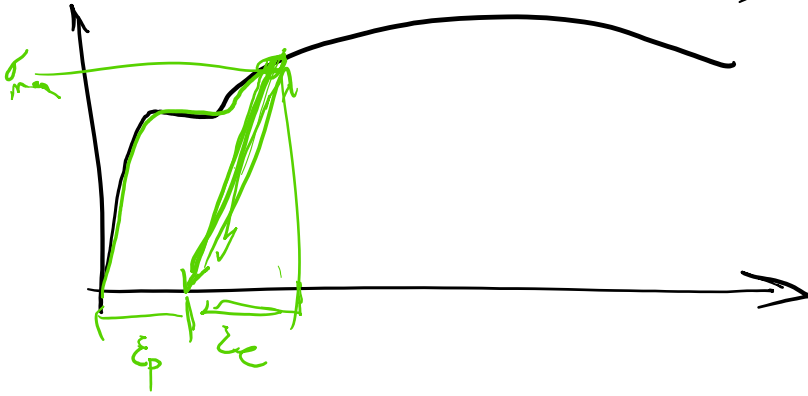
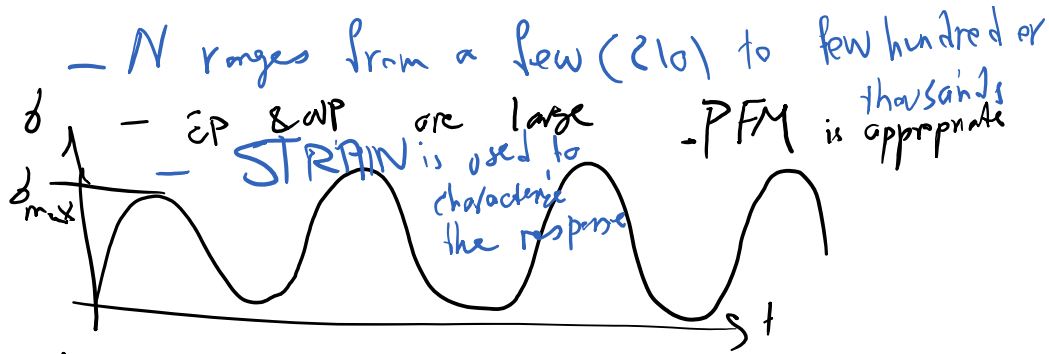
We only discuss this

	N	σ_{max}	$\Delta \epsilon^p / \Delta \epsilon^e$	$\Delta W^p / \Delta W^e$
Low cycle fatigue	10^2 to 10^4	σ_Y to σ_U	1 to 10	1 to 10
Very low cycle fatigue	1 to 20	$\approx \sigma_U$	10 to 100	10 to 100

Source: Dufailly and Lemaitre (1995)

LCF & V LCF

σ_{max} can get to or exceed σ_y

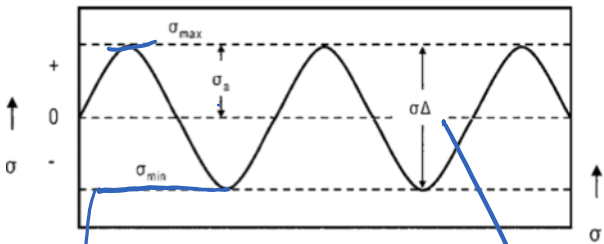


Some terminology:

Base line fatigue loading

$$\left. \begin{matrix} \sigma_{min} \\ \sigma_{max} \end{matrix} \right\} \rightarrow$$

$$\sigma_{max} = -\sigma_{min}$$



$$\Delta\sigma = \sigma_{max} - \sigma_{min}$$

$$\sigma_a = \frac{\Delta\sigma}{2} \quad (\text{for the figure on the right})$$

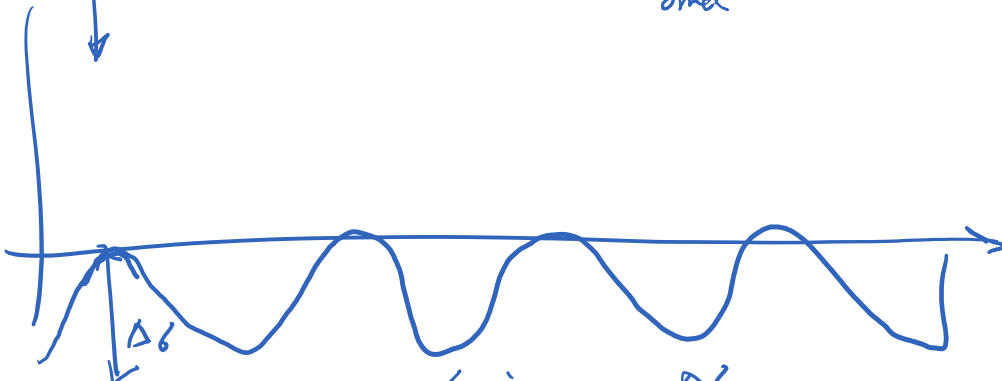
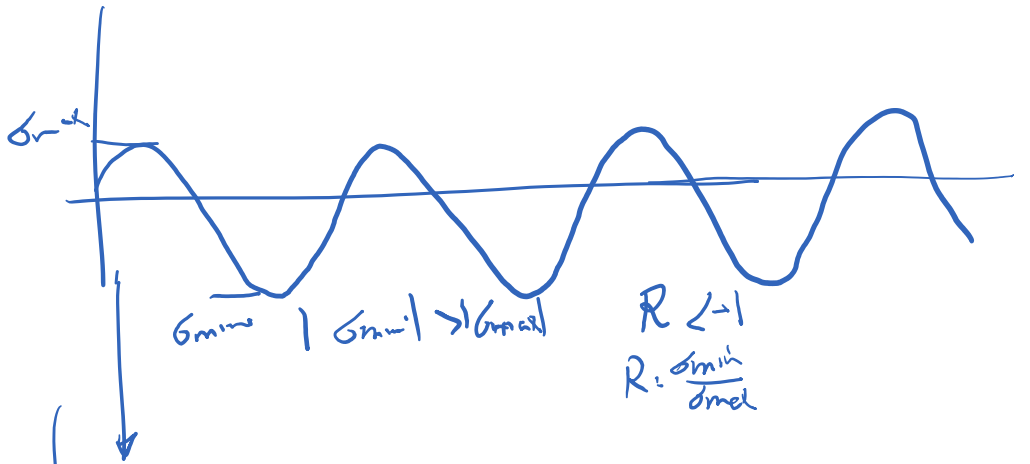
$$G_m = \frac{\sigma_{min} + \sigma_{max}}{2} \quad \sigma_a = \sigma_{max} \Rightarrow G_m = 0$$

$$R = \frac{\sigma_{min}}{\sigma_{max}}$$

$$R = -1$$

$$R \in (-\infty, 1)$$

make the loading more compressive

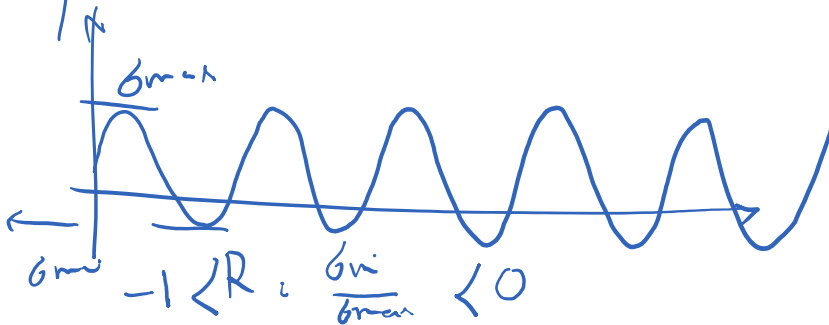


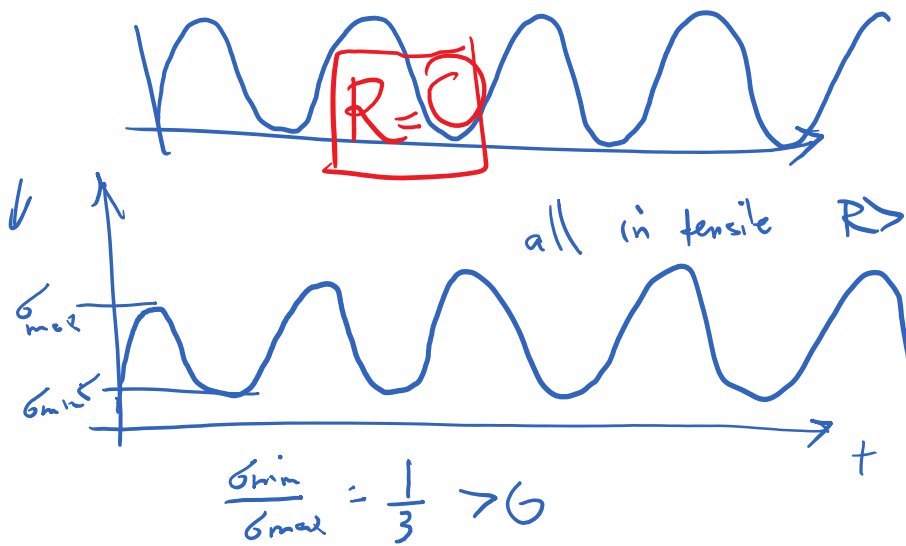
$$R = \frac{\sigma_{min}}{\sigma_{max}} \rightarrow \frac{-\Delta\sigma}{\sigma} \rightarrow -\infty$$

$R \rightarrow -\infty$ all compressive loading

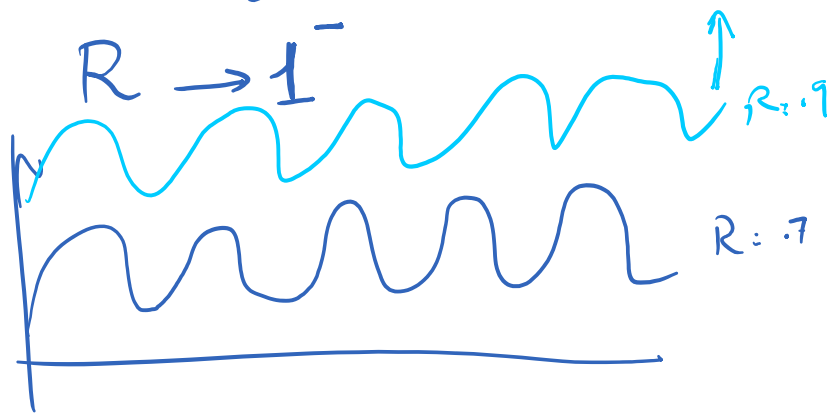
Let's make it more tensile:

$$\sigma_{max} = -\sigma_{min}$$





for a fixed $\Delta\sigma$ as $\sigma_m \uparrow$



$$R \in (-\infty, 1)$$

$$\Delta\sigma = \sigma_{\max} - \sigma_{\min}$$

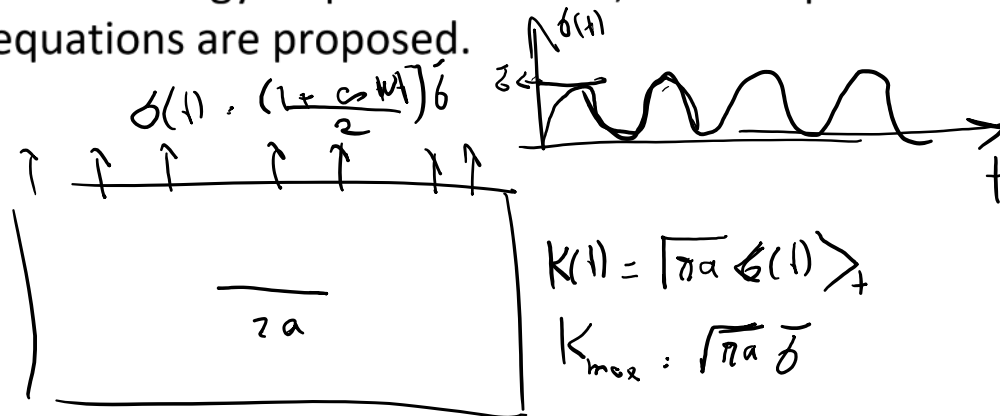
$$\sigma_a = 0.5(\sigma_{\max} - \sigma_{\min})$$

$$\sigma_m = 0.5(\sigma_{\max} + \sigma_{\min})$$

$$R = \frac{\sigma_{\min}}{\sigma_{\max}} \quad \text{load ratio}$$

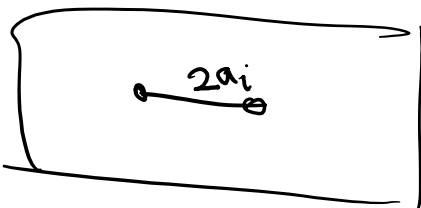
Cyclic vs. static loadings

- Static: Until K reaches K_c , crack will not grow
- Cyclic: K applied can be well below K_c , crack still grows!!!
- 1961, Paris et al used the theory of LEFM to explain fatigue cracking successfully.
- Methodology: experiments first, then empirical equations are proposed.



Compare K_{max} with K_{Ic}

Even with $K_{max} \ll K_{Ic}$ the crack can still grow very slowly

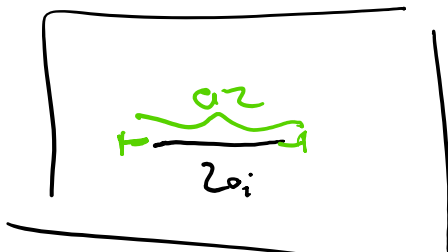


stage 1

$$\Delta \delta = \bar{\delta}$$

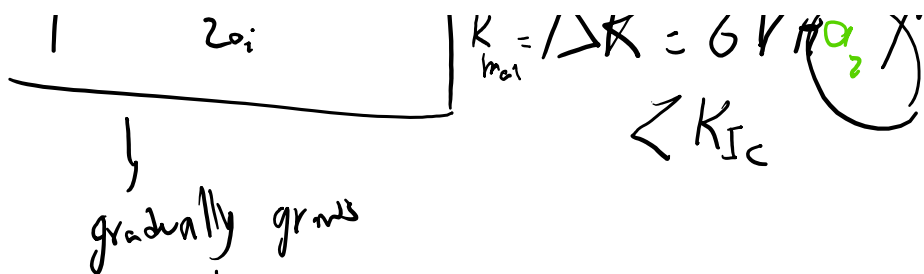
$$\Delta K = \bar{\delta} \sqrt{\pi a_i}$$

later time

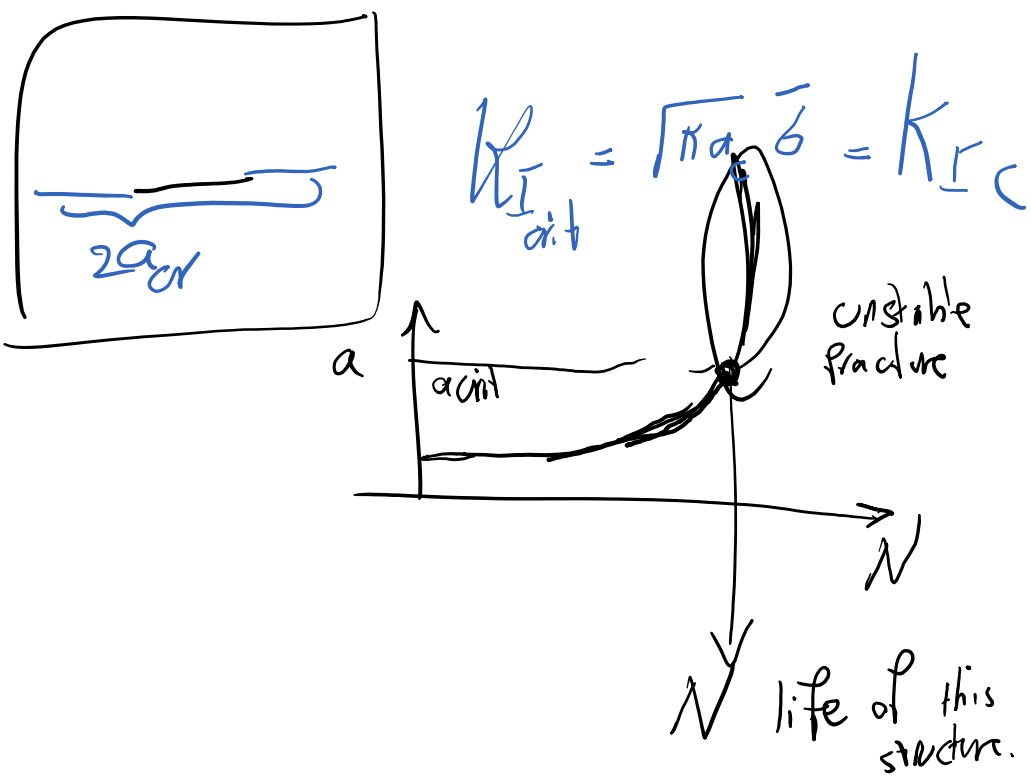


$$\Delta \delta = \bar{\delta} \text{ const}$$

$$K_{max} = \Delta K = \bar{\delta} \sqrt{\pi a_2}$$



$\cup \cap \text{TTT} \perp$ a is large enough
 that the crack can propagate in an
 unstable manner



Loading, geometry, ... $\longrightarrow N = ?$

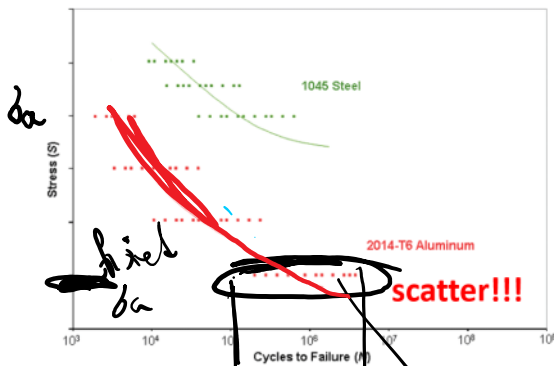
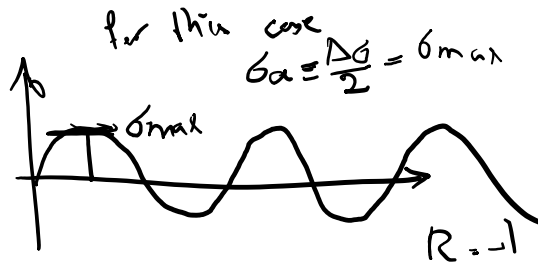
old approach	New approach
— Stress-based ($\Delta\sigma$)	Paris law
— Doesn't directly	— K -based

- Doesn't directly incorporate initial material flaws

- K-based
- we start with an initial crack length

S-N plots, S-N-P plots

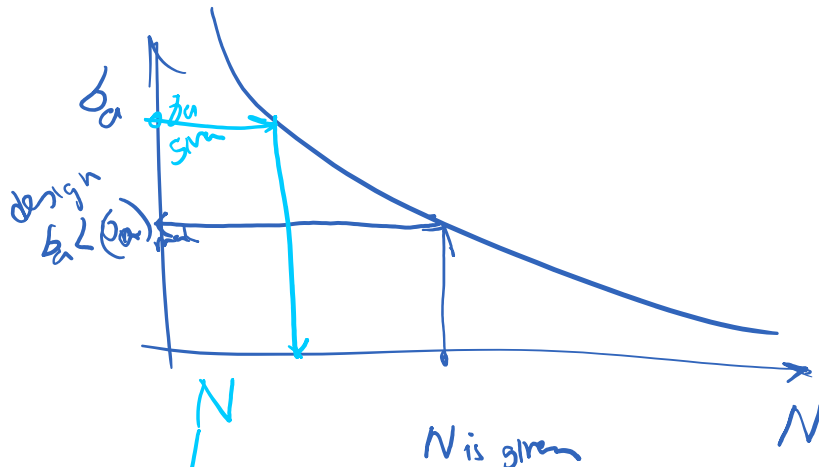
for R=1
or

10⁵
~ 5 x 10⁶

we observe a lot of scatter in N

greater than order of magnitude variation



specify the life of structure + helps

ii monitoring time intervals - to check the structure

