K-G relationship (cont.)

Mode I

$$\left| \begin{array}{c} G_I = \begin{cases} \frac{K_I^2}{E} & \text{plane stress} \\ (1-v^2) \frac{K_I^2}{E} & \text{plane strain} \end{cases} \right.$$

Mixed mode

$$G = rac{K_I^2}{E'} + rac{K_{II}^2}{E'} + rac{K_{III}^2}{2\mu} igg) _{E'= igg\{ egin{array}{c} rac{E}{1-
u^2} & ext{for plane strain} \ E & ext{ for plane stress} \end{array} igg\}$$

- Equivalence of the strain energy release rate and SIF approach
- Mixed mode: G is scalar => mode contributions are additive
- Assumption: self-similar crack growth!!!

Self-similar crack growth: planar crack remains planar (da same direction ${\rm as}a$)

Key use: by having K's we can determine if the crack grows or not.

Imagine we just have mode I fracture:



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Side note related to last time

KTE ~ (f(a) VTA) &



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Dynamic fragmentation of brittle solids: a multi-scale model

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Plastic Fracture Mechanics (PFM)

- When can we use LEFM

5.2. Plastic zone models

- 1D Models: Irwin, Dugdale, and Barenbolt models



ETIM is a good

100 MIO W

When "> / 'p



Find r_p:

Plastic correction: 1st order approximation

y s yy





To get the actual Process zone size (PZS) = r_y or r_p , we need to solve the problem (BC + PDE + constitutive equations) from the beginning to find the correct solution. Where around the crack the material yields or undergoes large inelastic deformation is called the PZS.

Rather than doing this difficult problem, we can do a bit better than the 1st order approximation as shown below:



This is better than r_y, but still is not doing any stress redistribution.



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$$\begin{aligned} \int G_{1} & G_{2} &$$

$$\frac{1}{1-2\nu} = \frac{1}{1-2\nu} \frac{1}{1-$$