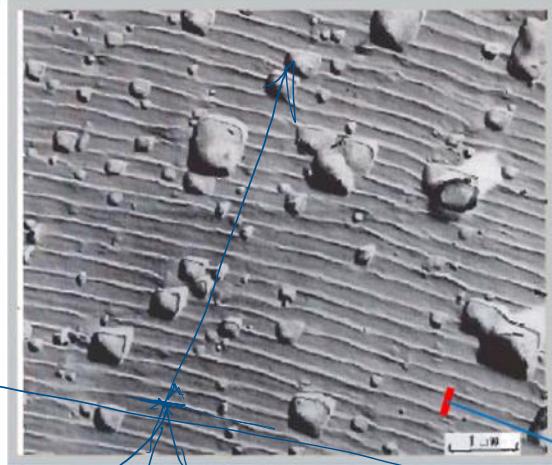


Fatigue striations

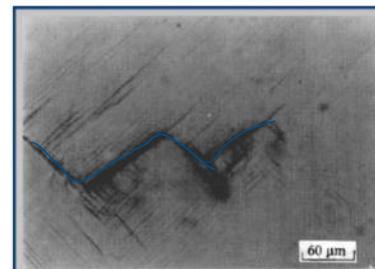
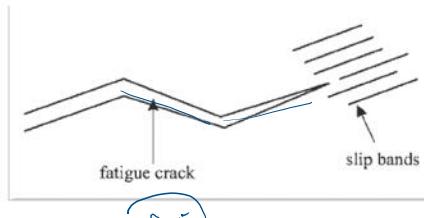


Fracture surface of a 2024-T3 aluminum alloy
(source S. Suresh MIT)

Striation caused by individual microscale
crack advance incidents

Fatigue crack growth:

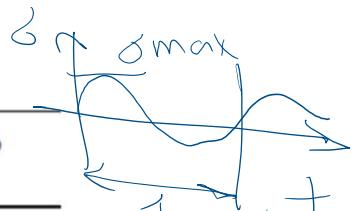
Microcrack formation in **accumulated slip bands** due to repeated loading



Fatigue Regimes

Table 7.1 Classification of fatigue damage

Fatigue	Failure cycles N_f	Pertinent stress	Strain ratio $\Delta \varepsilon^p / \Delta \varepsilon^e$	Energy ratio $\Delta W_p / \Delta W_e$
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Fatigue	Failure cycles N_R	Pertinent stress	Strain ratio $\Delta\varepsilon^p/\Delta\varepsilon^e$	Energy ratio $\Delta W^p/\Delta W^e$
Very high cycle fatigue	$> 10^7$	$< \sigma_F$	≈ 0	≈ 0
High cycle fatigue	10^5 to 10^6	$< \sigma_Y$	≈ 0	≈ 0
Low cycle fatigue	10^2 to 10^4	σ_Y to σ_U	1 to 10	1 to 10
Very low cycle fatigue	1 to 20	$\approx \sigma_U$	10 to 100	10 to 100

Source: Dufailly and Lemaître (1995)

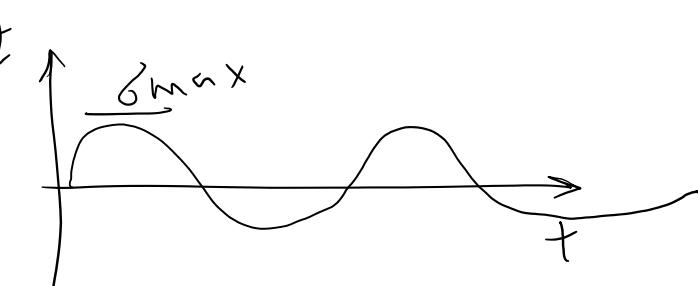
• **Very high cycle and high cycle fatigue:**

- Stresses are well below yield/ultimate strength.
- There is almost no plastic deformation (in terms of strain and energy ratios)
- Fatigue models based on LEFM theory (e.g. SIF K) are applicable.
- Stress-life approaches are used (stress-centered criteria)

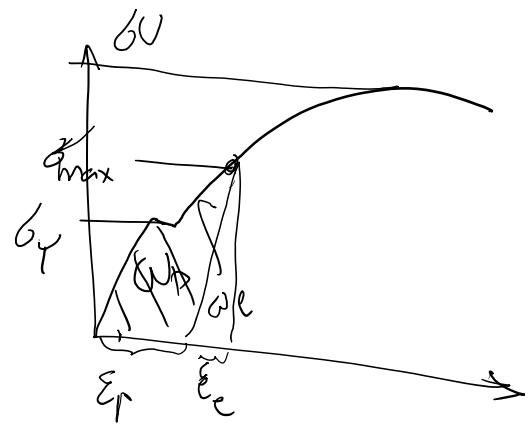
• **Low cycle and very low cycle fatigue:**

- Stresses are in the order of yield/ultimate strength.
- There is considerable plastic deformation.
- Fatigue models based on PFM theory (e.g. J integral) are applicable.
- Strain-life approaches are used (strain-centered criteria)

{ LEFM ✓
- stress based

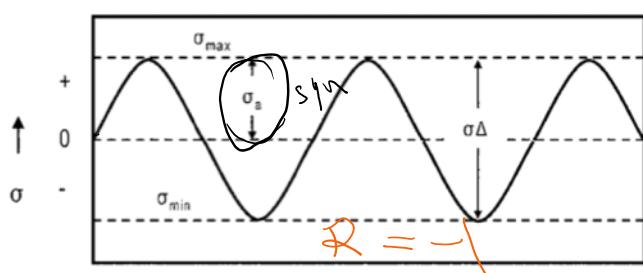


{ PFM,...
strain -driven

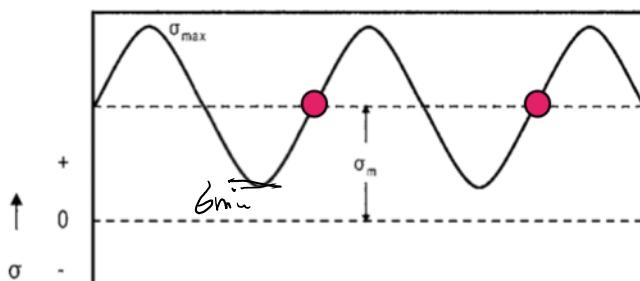


(not covered here)

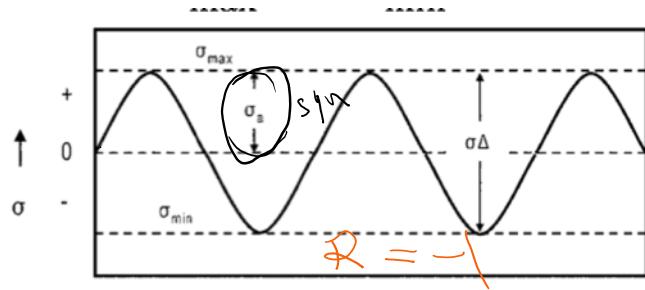
Definitions from a cyclic loading:



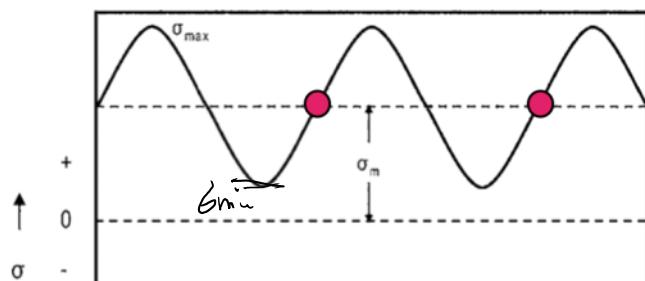
Fully Reversed Loading



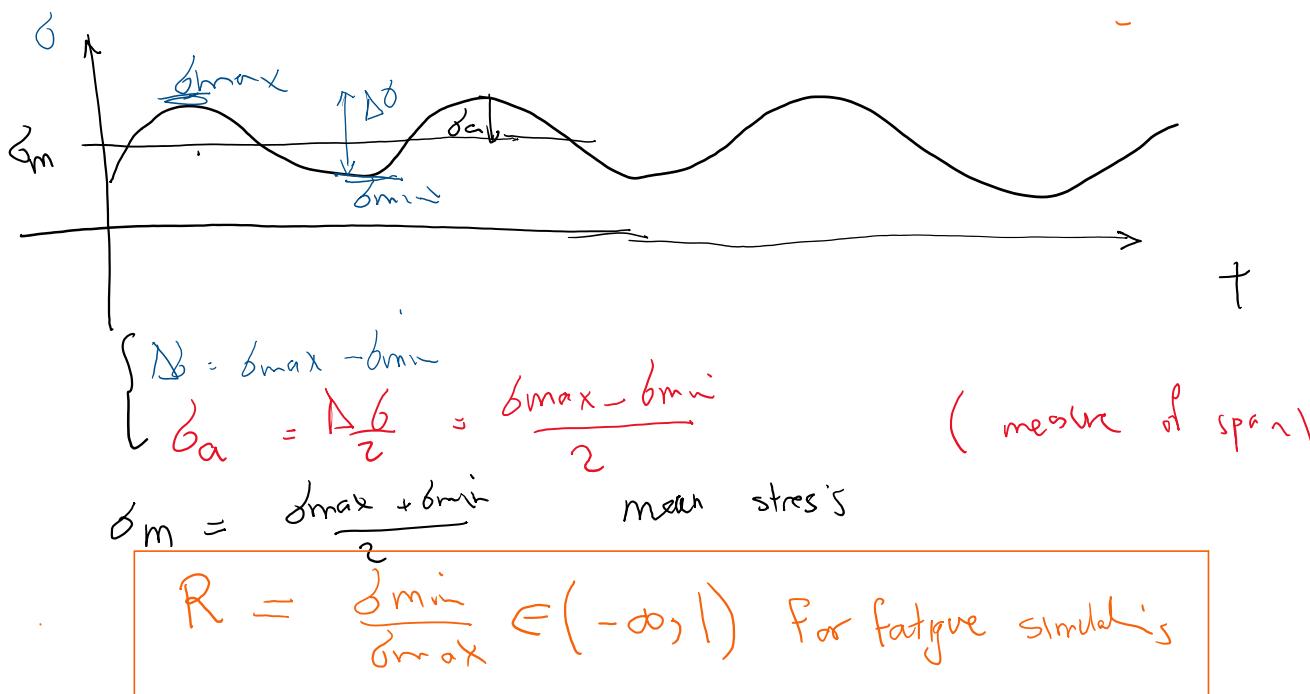
Tension-Tension with Applied Stress



Fully Reversed Loading

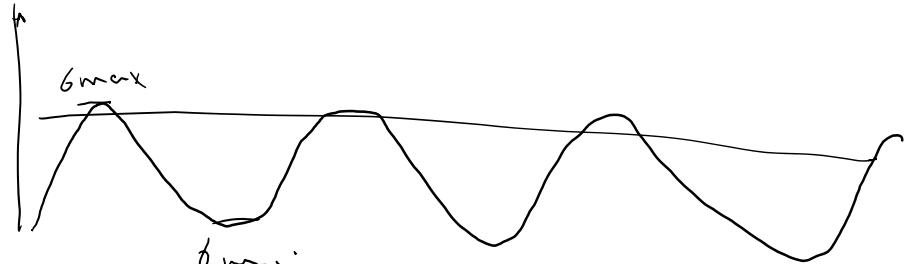


Tension-Tension with Applied Stress



lowest R

$$R = \frac{\delta_{min}}{\delta_{max}}$$



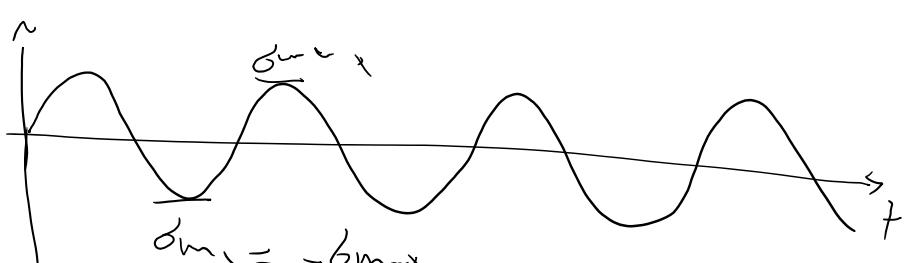
best case

$R \rightarrow -\infty$

$\rightarrow 1$

$R = -1$

many experiments done

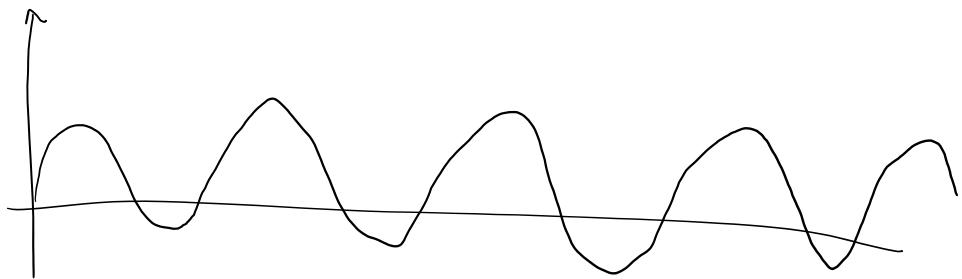


$$R = -1$$

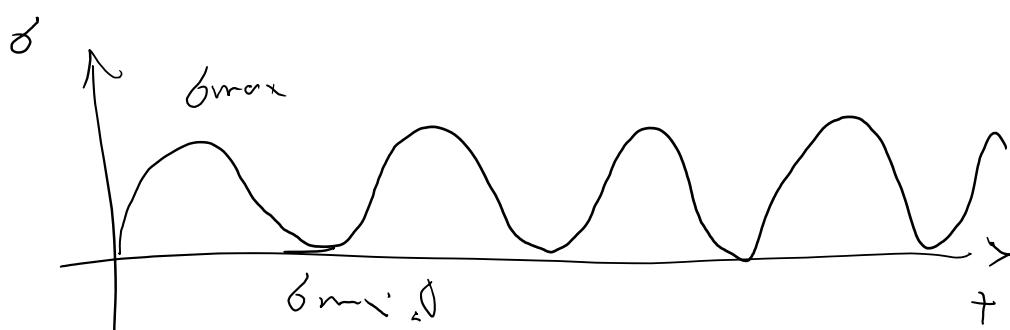
many experiments done like this

$$\sigma_m = -\delta_{\max}$$

$$-1 < R < 0$$



$$R = \frac{\delta_{\min}}{\delta_{\max}} = 0$$



$$R > 0$$

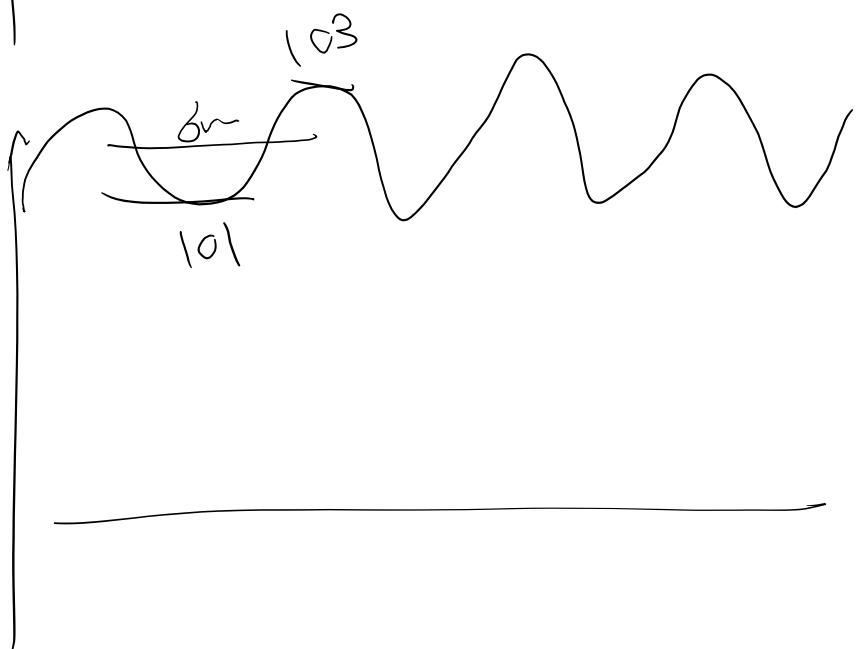
$$R = \frac{1}{3} > 0$$



$$R = \frac{101}{103} \approx 1^-$$

$$\text{as } \delta_m \rightarrow \infty$$

$$R \rightarrow 1^-$$



$$\Delta\sigma = \sigma_{\max} - \sigma_{\min}$$

$$\sigma_a = 0.5(\sigma_{\max} - \sigma_{\min})$$

$$\sigma_m = 0.5(\sigma_{\max} + \sigma_{\min})$$

$$R = \frac{\sigma_{\min}}{\sigma_{\max}}$$

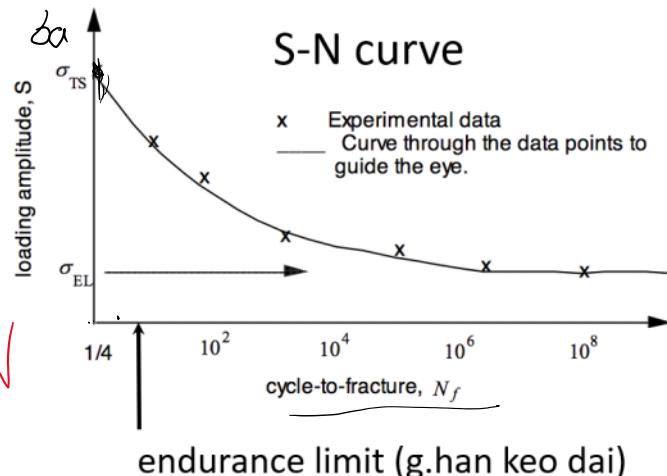
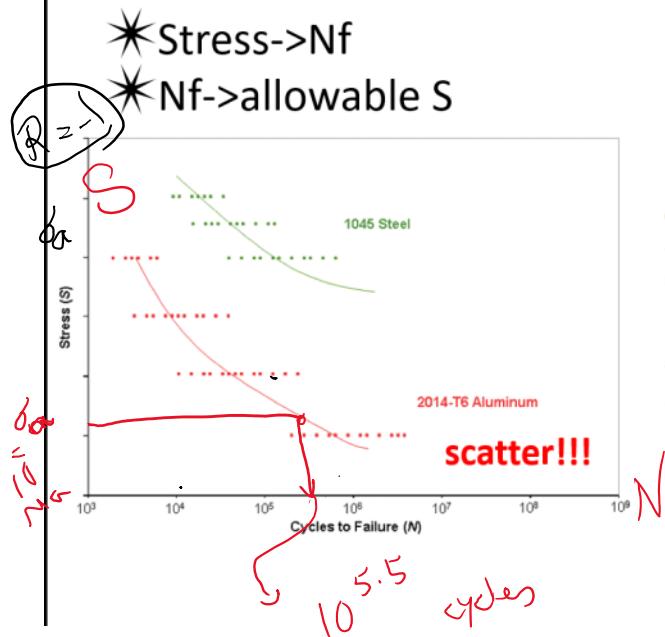
load ratio

Two approaches for fatigue:

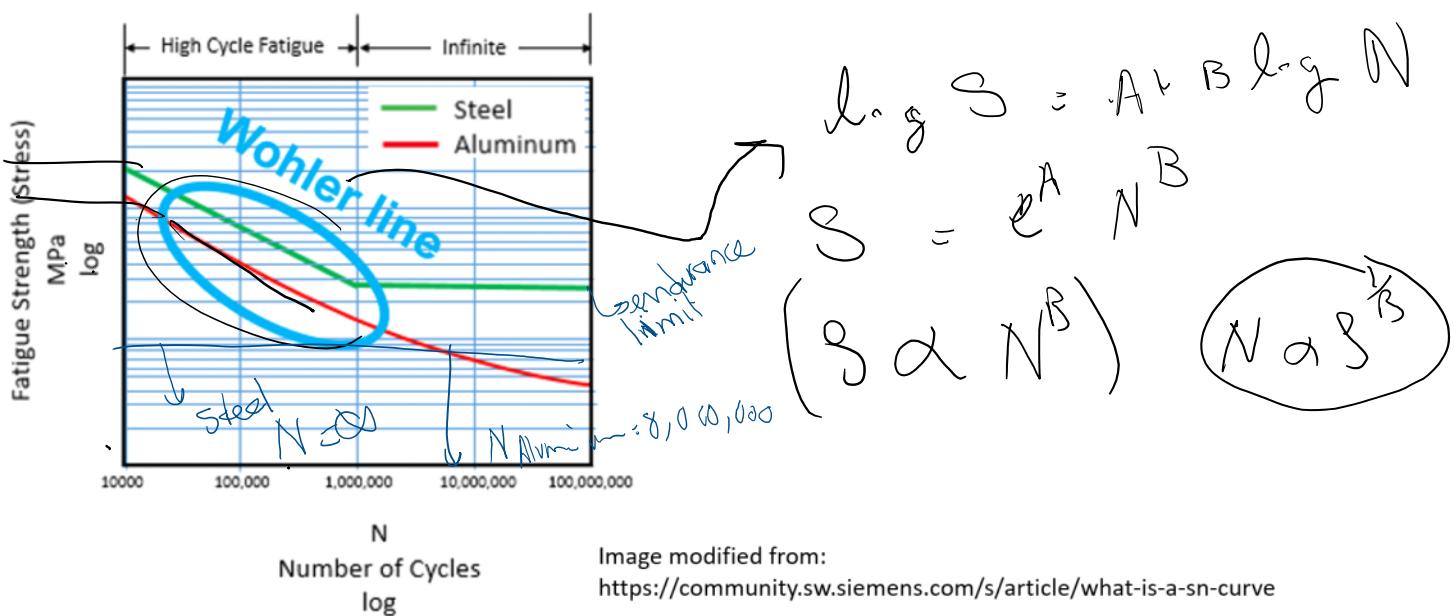
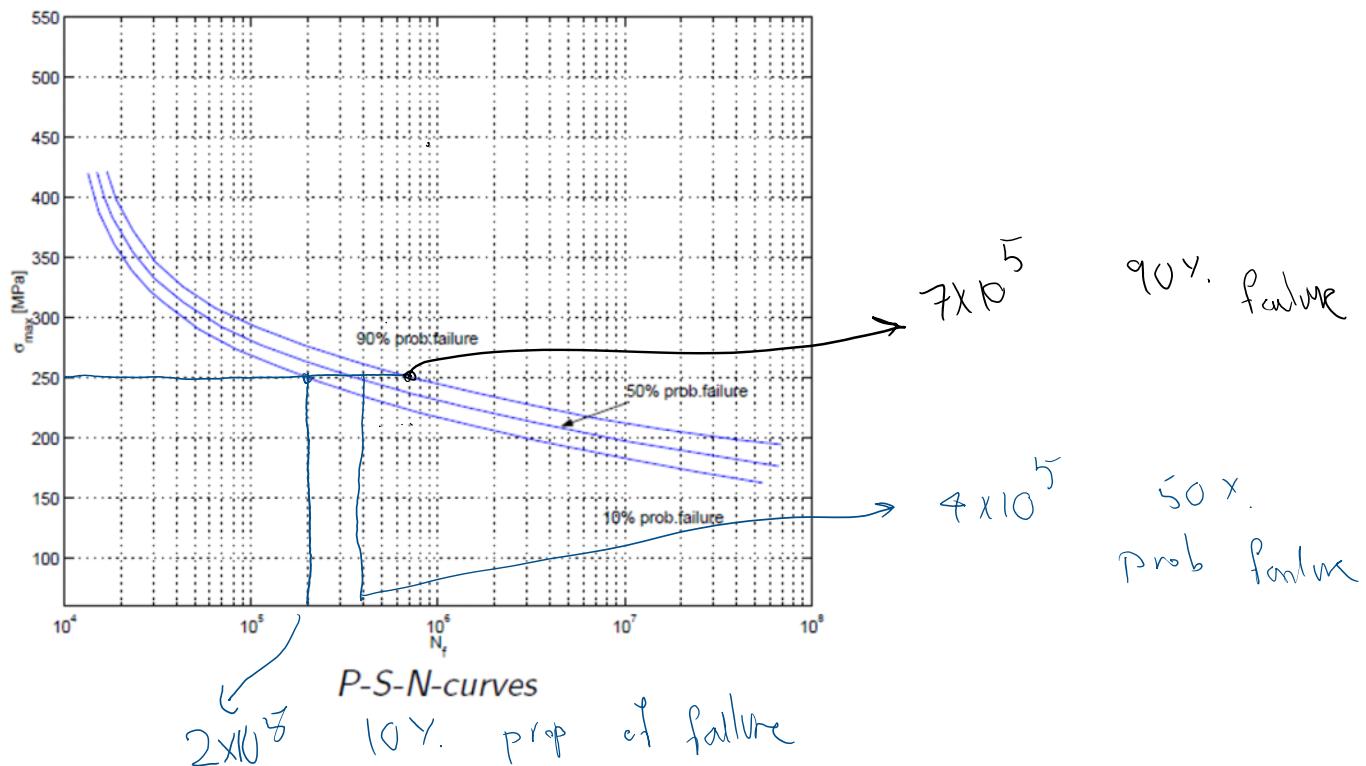
- 1) Older one: S-N plots
- 2) Paris-Erdogan relation

S-N curve

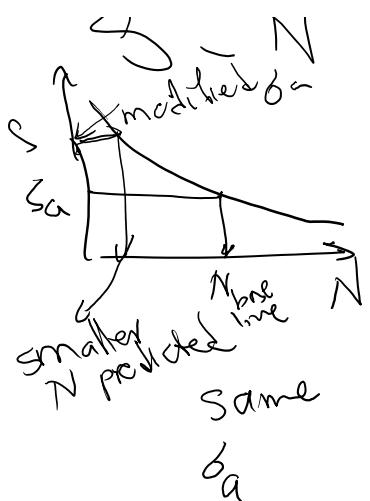
Reminder: [ASTM](#) defines *fatigue life*, N_f , as the number of stress cycles of a specified character that a specimen sustains before [failure](#) of a specified nature occurs.



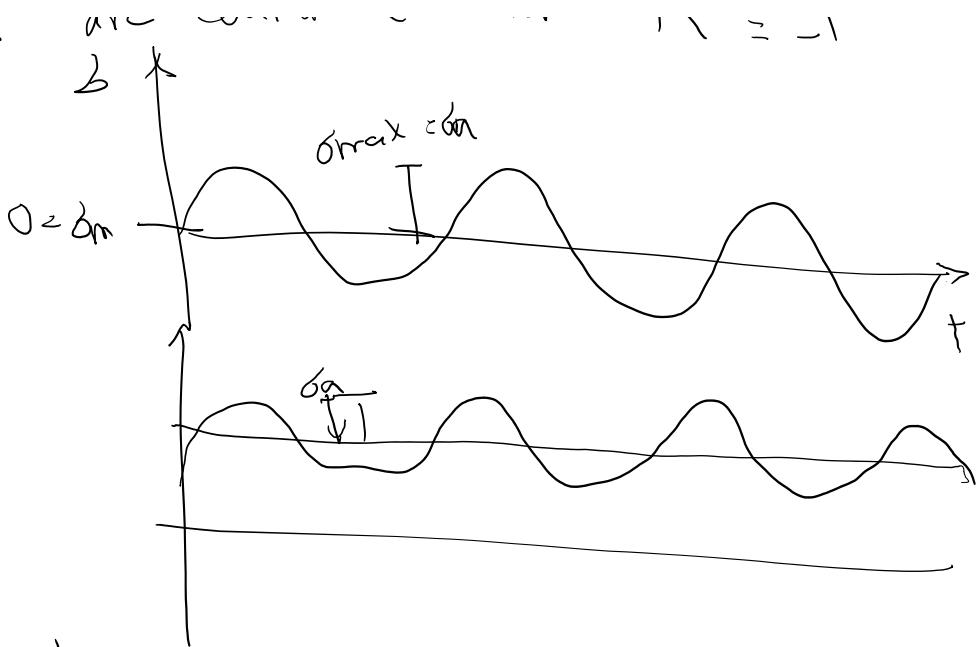
S-N-P curve: scatter effects



σ - N plots are calibrated for $R = -1$



plots

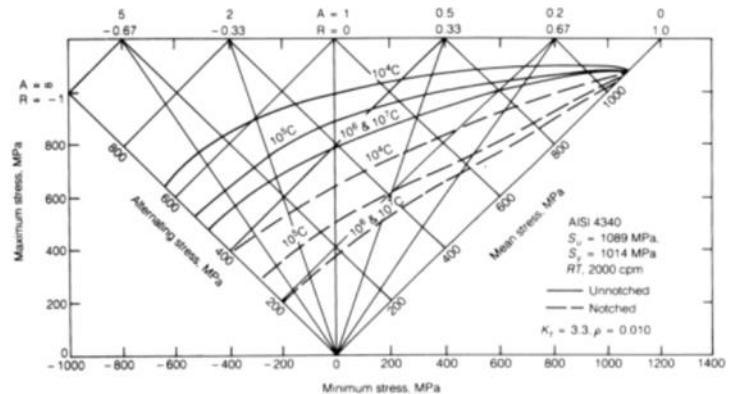
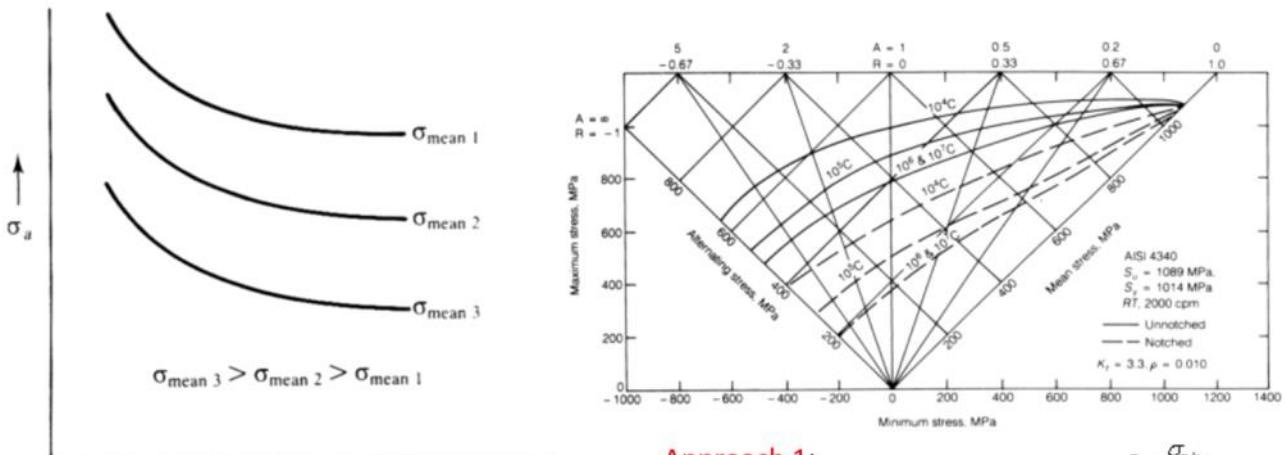


We get fewer cycles now

δ_a
 we'll
 take
 original S-N

$$\sigma_a = \sigma_{f0} \left[1 - \left(\frac{\sigma_m}{\sigma_u} \right)^y \right]$$

Effect of mean stress



Approach 2:
Correction-factor formulas

$$\sigma_a = \sigma_{f0} \left[1 - \left(\frac{\sigma_m}{\sigma_u} \right)^r \right]$$

where σ_a is the amplitude of allowable stress (alternating stress).

σ_{f0} is the stress at fatigue fracture when the material under zero mean stress cycled loading

σ_m is the mean stress of the actual loading.

σ_u is the tensile strength of the material.

$r = 1$ is called Goodman line which is close to the results of notched specimens.

$r = 2$ is the Gerber parabola which better represents ductile metals.

Approach 1:
Master diagram

$$R = \frac{\sigma_{m,n}}{\sigma_{\max}}$$

$$A = \frac{\sigma_a}{\sigma_m} = \frac{1-R}{1+R}$$

Other correction factor

Gerber (1874)

$$\frac{\sigma_a^*}{\sigma_a} = 1 - \left(\frac{\sigma_m}{\sigma_u} \right)^2$$

Goodman (1899)

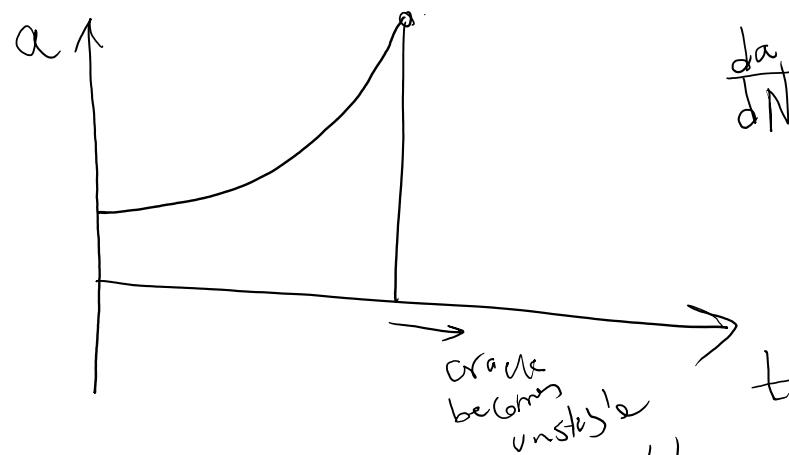
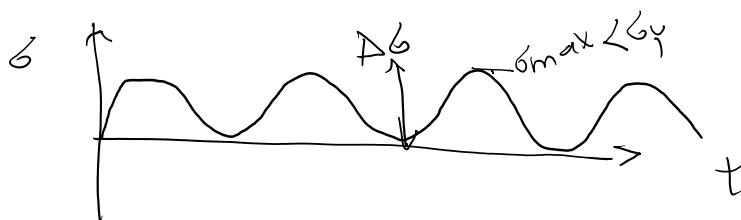
$$\frac{\sigma_a^*}{\sigma_a} = 1 - \frac{\sigma_m}{\sigma_u}$$

Soderberg (1939)

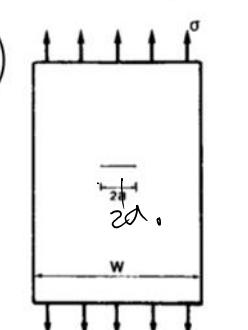
$$\frac{\sigma_a^*}{\sigma_a} = 1 - \frac{\sigma_m}{\sigma_{y0}}$$

Approach 2) Paris-Erdogan

$$R = 0$$



$\frac{da}{dN}$
rate of
crack growth



$$\frac{da}{dt}$$

$$\propto \Delta \delta$$

crack
below
unstable
 N

t

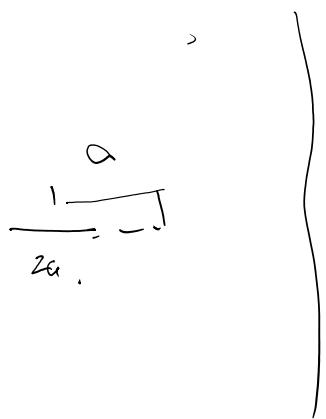
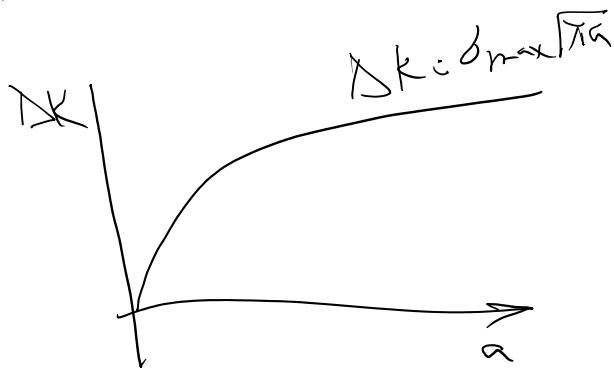
$$K_{max} = \sigma_{max} \sqrt{\pi a}$$

$$K_m = \sigma_m \sqrt{\pi a} = 0$$

assume an infinite domain

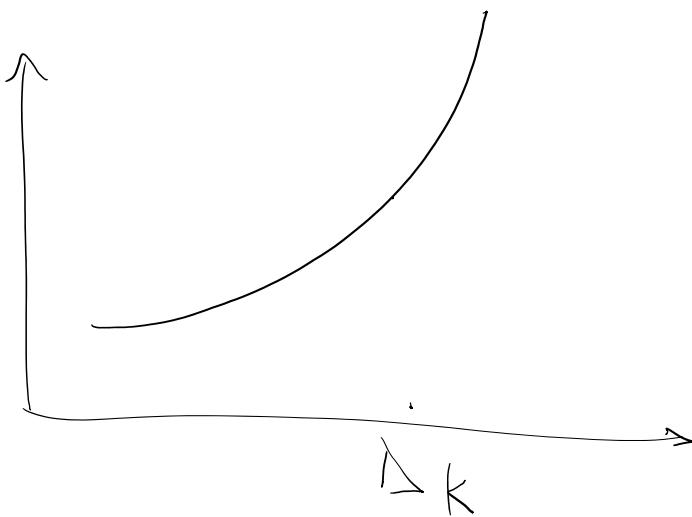
$$\Delta K = \left(\sigma_{max} \sqrt{\pi a} \right)$$

how does it behave versus a

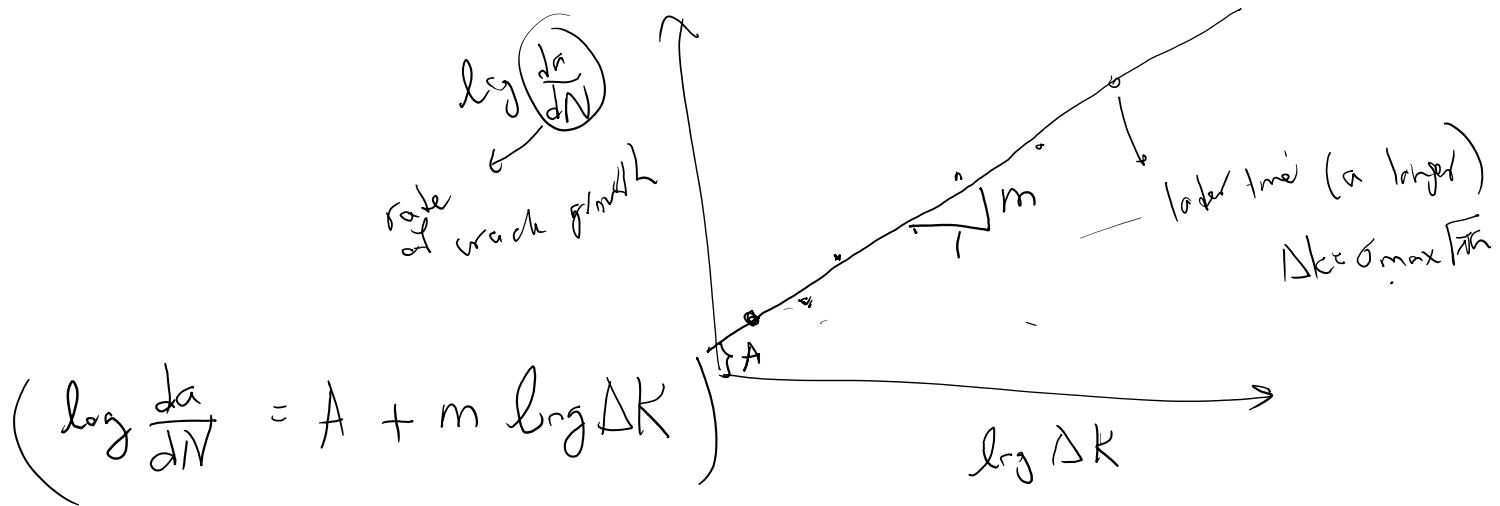


as the crack grows

$$\frac{da}{dN}$$



apply log on both axes



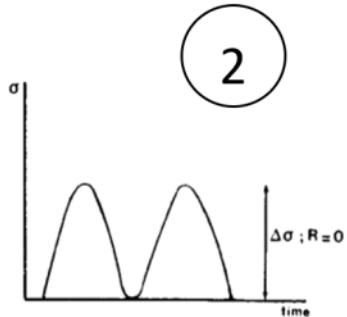
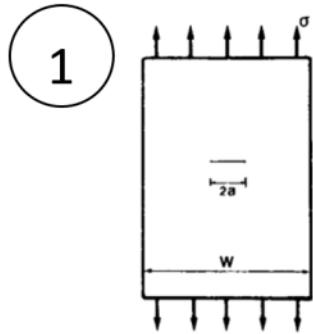
$$\frac{da}{dN} = C(e^A) \Delta K^m$$

$\frac{da}{dN} = C \Delta K^m$

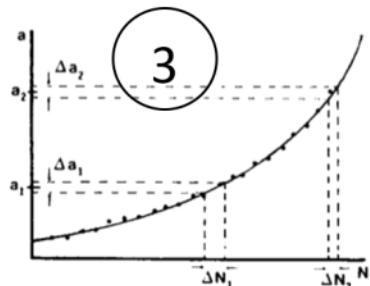
Paris - Erdogan Relation
 needs an initial crack

Crack growth data

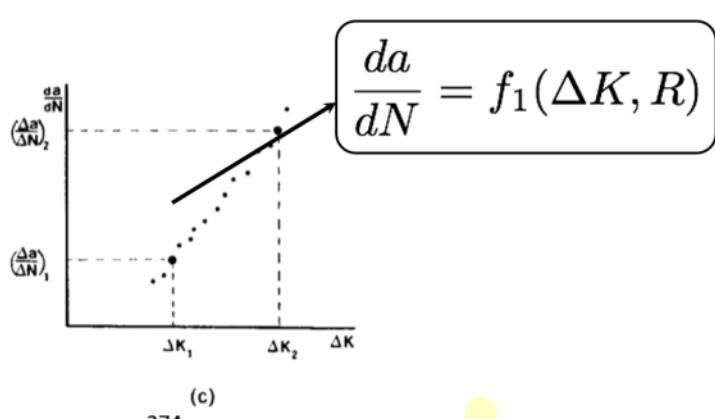
$$K = \sigma \sqrt{\pi a}$$



(a)

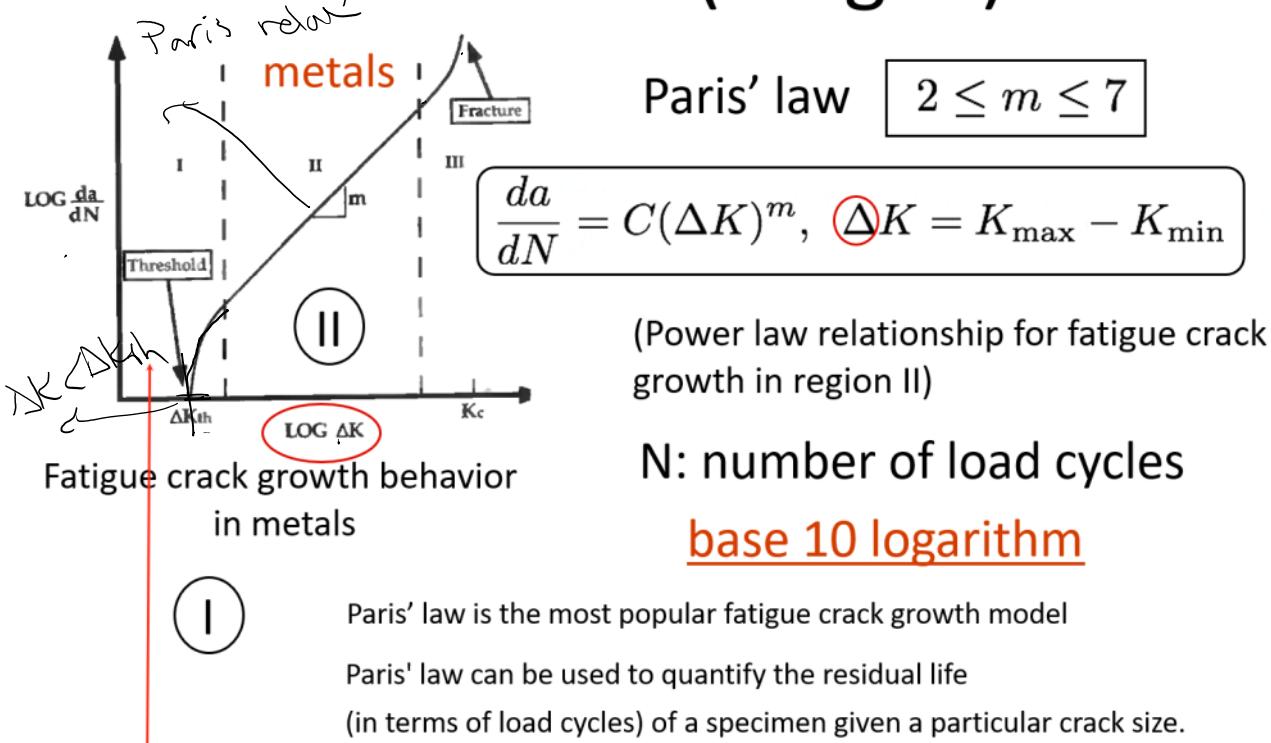


(b)



(c)

Paris' law (fatigue)



$\Delta K \leq \Delta K_{th}$: no crack growth
(dormant period) 10^{-8} mm/cycle

375

not depends on load ratio R

$$\frac{da}{dN} = C(\Delta K)^m, \Delta K = K_{\max} - K_{\min}$$

Table 1: Numerical parameters in the Paris equation.

alloy	m	A
Steel	3	10^{-11}
Aluminum	3	10^{-12}
Nickel	3.3	4×10^{-12}
Titanium	5	10^{-11}

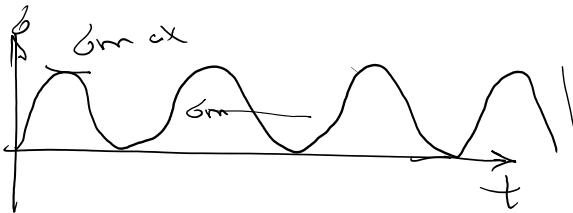
C, m

are material properties that must be determined experimentally from a $\log(\Delta K)$ - $\log(da/dN)$ plot.

m
2-4 metals
4-100 ceramics/ polymers

6mm ax

infinite domain



infinite domain

$$\Delta K = \delta_{\max} \sqrt{\pi a}$$

$$\begin{aligned}\frac{da}{dN} &= C \Delta K^m \\ &= C \left(\delta_{\max} \sqrt{\pi a} \right)^m \\ \frac{da}{dN} &= \left(C \delta_{\max}^m \pi^{\frac{m}{2}} a \right) \text{ red } \\ A &= C \delta_{\max}^m \pi^{\frac{m}{2}}\end{aligned}$$

$$2a = 2a_i \quad \text{uniformly}$$

$$\frac{da}{dN} = A a^{\frac{m}{2}} \rightarrow \frac{da}{a^{\frac{m}{2}}} = A dN$$

$$a^{-\frac{m}{2}} da = A dN \quad \text{integrate}$$

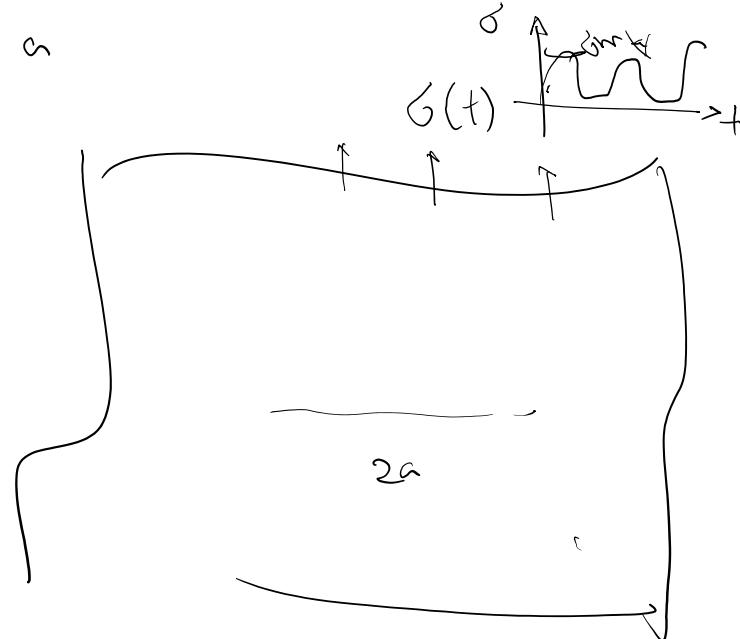
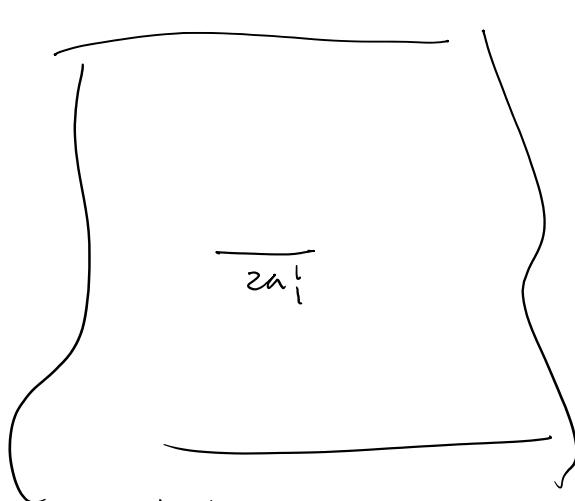
$$\int_{a_i}^a a^{-\frac{m}{2}} da = \int_0^N A dN \Rightarrow \frac{1}{1 - \frac{m}{2}} a^{1 - \frac{m}{2}} \Big|_{a_i}^a = AN(a)$$

$$\Rightarrow \frac{1}{\frac{m}{2} - 1} \frac{1}{a_i^{\frac{m}{2}-1}} - \frac{1}{\frac{m}{2} - 1} \frac{1}{a^{\frac{m}{2}-1}} = AN(a)$$

$$(2) N(a) = \frac{1}{\left(\frac{m}{2}\right) C \pi^{\frac{m}{2}} (\Delta \delta)^m} \left(\frac{1}{a_i^{\frac{m}{2}-1}} - \frac{1}{a^{\frac{m}{2}-1}} \right)^{\frac{m}{2}}$$

m>2

cycles to reach crack length a



$$K_{max} = \sigma_{max} \sqrt{\pi a}$$

sudden fracture when

We'll end up with
 $K_{max} = K_{Ic}$

$$\sigma_{max} \sqrt{\pi a_f} = K_{Ic} \Rightarrow$$

$\frac{a_f}{\text{force}}$

$$a_f = \frac{1}{\pi} \left(\frac{K_{Ic}}{\sigma_{max}} \right)^2 \quad (3)$$

$$N(a) = \frac{1}{(\frac{m}{2}-1) \pi^{\frac{m}{2}} \Delta \sigma^{\frac{m}{2}}} \left(\frac{1}{a_i^{\frac{m}{2}}} - \frac{1}{a^{\frac{m}{2}}} \right) \quad (2)$$

$\overbrace{-}^{2a_i}$ How long does it take to fail?

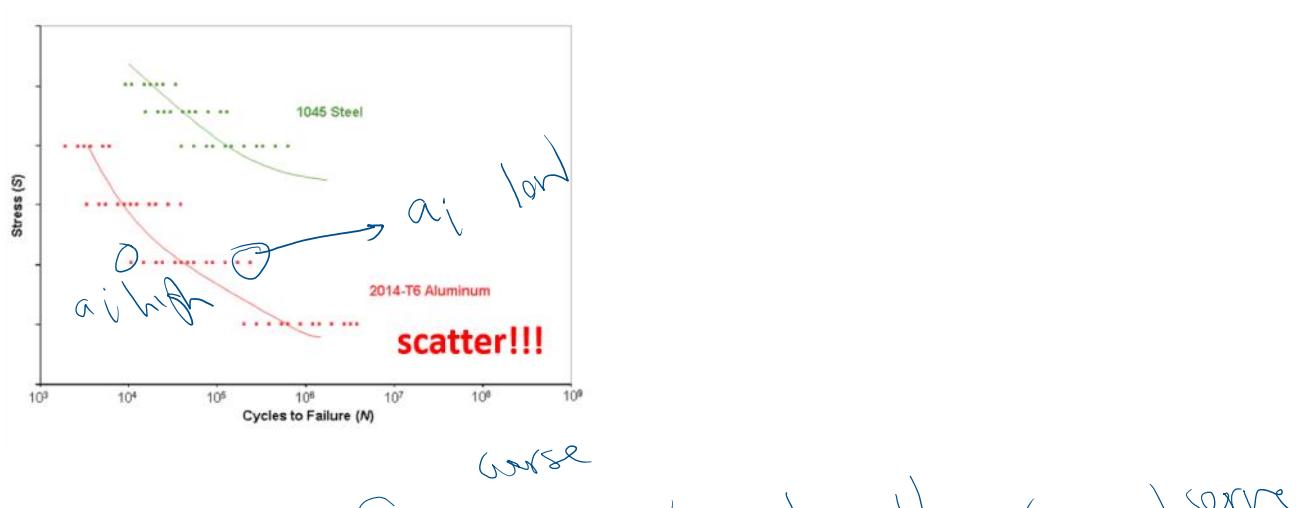
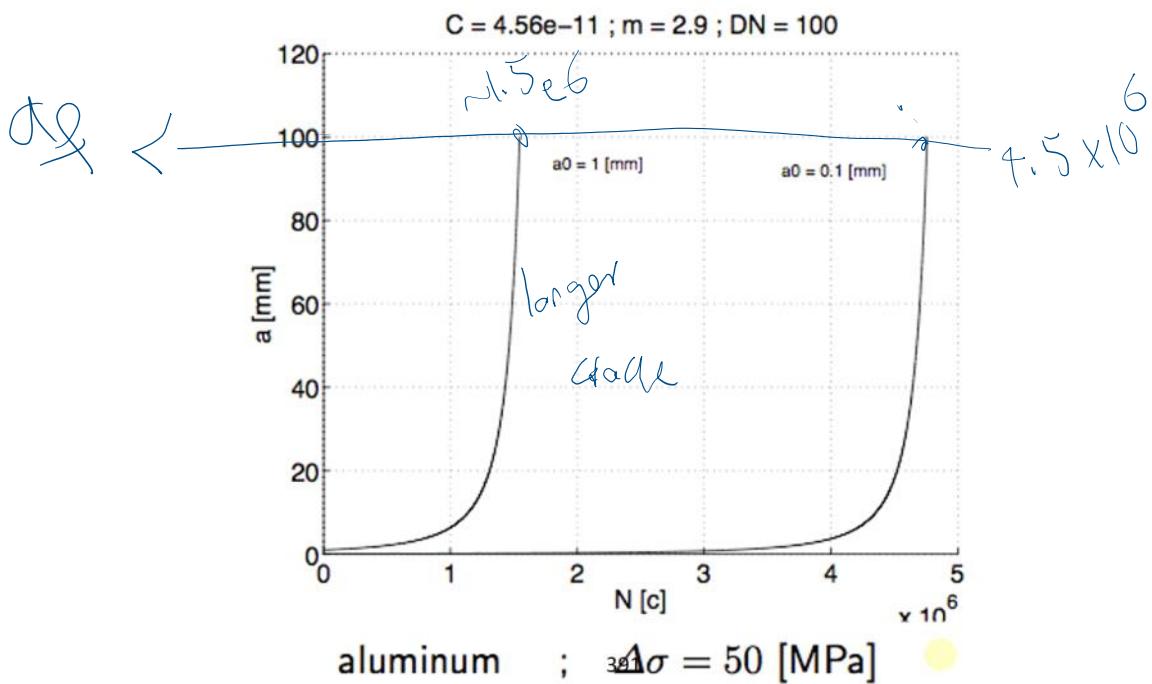
$$N_f(a_i) = \frac{1}{(\frac{m}{2}-1) \pi^{\frac{m}{2}} \Delta \sigma^{\frac{m}{2}}} \left(\frac{1}{a_i^{\frac{m}{2}}} - \frac{1}{a_f^{\frac{m}{2}}} \right)$$

(4) 1, 2

(4)

$$\alpha_f = \frac{1}{\pi} \left(\frac{K_{Ic}}{\sigma_{max}} \right)^2$$

Importance of initial crack length



$$a_i \approx$$

worst crack length we observe
at a_{tol} the measurement tolerance